Electrical Properties of CdFeSe, CdMnTe Epitaxial Films

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Abstract: - Optimal conditions of obtaining perfect $Cd_{1-x}Fe_xSe$ (x<0.08) and $Cd_{1-x}Mn_xTe$ (x=0.15) epitaxial films are defined. The electrical properties of $Cd_{1-x}Fe_xSe$ (x<0.08) and $Cd_{1-x}Mn_xTe$ (x=0.15) epitaxial films have been studied at room temperature. It was defined that $Cd_{1-x}Fe_xSe$ semimagnetic semiconductor epitaxial films are of n-type and $Cd_{1-x}Mn_xTe$ p-type. Electrical resistivity was defined 14.4 · 10⁷ Ohm·cm. The effect of γ-irradiation on VAC of $Cd_{1-x}Mn_xTe$ epitaxial films is studied at doses $D_y \le 1.5$ κGy.

Key-Words: - Semimagnetic semiconductors, molecular beam condensation, XRD, SEM, electrical, VAC

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1 Introduction

Semimagnetic semiconductors (SMSC) are of II-VI and IV-VI group compounds in which a fraction of nonmagnetic cations has been substituted by magnetic transition metal ions. Most of the research performed so far on these materials has been devoted to Mn-based SMSC, which represents a rather simple magnetic case, since the Mn^{+2} ions possess only spin momentum (S = 5/2, L = 0) [1], [2], [3], [4].

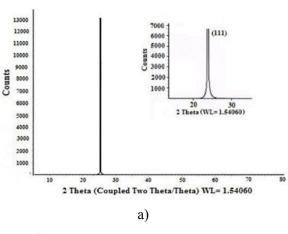
development natural of growing crystals with other transition metals, in particular, with Fe⁺². Materials based on Fe are not a simple extension of the SMSC family, since the physical situation, in this case, differs completely from that of Mn: substitutional Fe⁺² possesses both spin and orbital momenta (S = 2, L= 2). The problem of the Fe⁺² dopant in II-VI compounds has been studied for a long time. Unlike the well-known Mn-based DMS [5], the Fe-based II-VI compounds show a very low solubility of Fe, creating some difficulties in obtaining single - phase samples for studying the electronic properties. The $Cd_{1-x}Fe_xSe$ DMS are singlephase for x < 0.15. In $Cd_{1-x}Fe_xSe$ the transition metal Fe²⁺ ions replace randomly the Cd cations in the wurtzite structure of the host CdSe crystal [2]. Similarly to the Mn-based DMS's [6], [7], there is a large exchange interaction between the Fe 3d electrons, which leads to the spin splitting of the Fe 3d states into two groups of sublevels (spin-up and spin-down). In this work the conditions of obtaining perfect Cd_{1-x}Fe_xSe (x < 0.08) and Cd_{1-x}Mn_xTe (x = 0.15) epitaxial films

and study of their electrical properties have been studied.

2 Experimental

Thin films of $Cd_{1-x}Fe_xSe$ (x < 0.08) and $Cd_{1-x}Mn_xTe$ (x=0.15) epitaxial films were obtained by the Molecular Beam Condensation method in a vacuum of $(1\div2)10^{-4}$ Pa on glass substrates. It is determined the optimal conditions to obtain epitaxial films with perfect structure and a clean, smooth surface. The substrate temperature was $T_{sub}=640 \div 670$ K and source temperature was $T_{sub}=1100 \div 1200$ K. Epitaxial films were growing in the (111) plane of a face-centered cubic lattice.

Crystal structure of investigated epitaxial films was studied by X-ray diffraction (XRD) method on Bruker, Germany D8 ADVANGE X-ray diffractometer. XRD studies show that thin films grown on glass substrates at temperature $T_{sub} = 640 \div 670$ K have a monocrystalline (fig.1,a) and polycrystalline (fig.1,b) structures. To characterize a film quality, the full width at half maximum (FWHM) of diffraction peak is used. The FWHM of diffraction peak for $Cd_{1-x}Fe_xSe$ (x = 0.04) is $W_{1/2} = 1000-1100$ " (fig.1,a, insertion) and for $Cd_{1-x}Mn_xTe$ (x = 0.15) is $W_{1/2} = 1000-1100$ " (fig.1,b, insertion).



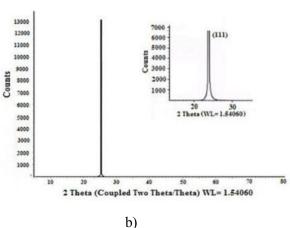
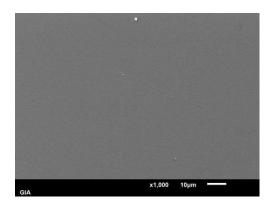


Fig. 1. X-ray diffraction pattern of a) $Cd_{1-x}Fe_xSe$ (x = 0.04), b) $Cd_{1-x}Mn_xTe$ (x = 0.15)

The surface morphology of investigated epitaxial films was studied by Scanning Electron Microscope (SEM) method on JEOL JSM-7600F Field Emission SEM (fig. 2). SEM image shows that the obtained Cd_{1-x}Fe_xSe epitaxial films were smooth and glossy (fig.2,a), but surface of Cd_{1-x}Mn_xTe epitaxial films were roughness with a grain size of up to 10 microns (fig. 2,b).



a)

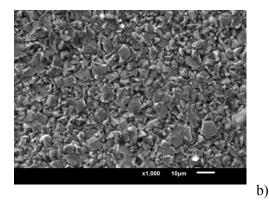


Fig. 2. SEM image of the a) $Cd_{1-x}Fe_xSe$ (x = 0.04) b) $Cd_{1-x}Mn_xTe$ (x = 0.15) epitaxial films obtained on the glass substrate

3 Results and Discussion

The dark electrical resistivity of $Cd_{1-x}Fe_xSe$, x < 0.08 epitaxial films was measured at T = 300 K temperature. To measure electrical resistivity silver paste was applied to make Ohmic contacts to $Cd_{1-x}Fe_xSe$, x < 0.08 epitaxial films. The nature of contact was checked up to 50 V. The VAC of $Cd_{1-x}Fe_xSe$, x < 0.08 epitaxial films are shown in fig. 3. The VAC has linear nature which confirms that silver produces Ohmic contact with $Cd_{1-x}Fe_xSe$. The measurement of thermo–emf across $Cd_{1-x}Fe_xSe$ SMSC epitaxial films confirms that films are of n–type.

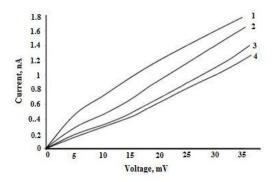


Fig. 3. VAC of Cd_{1-x}Fe_xSe epitaxial films: 1- CdSe, 2- Cd_{0.6}Fe_{0.4}Se, 3- Cd_{0.2}Fe_{0.8}Se, 4-FeSe

VAC of $Cd_{1-x}Mn_xTe$ (x = 0.15) epitaxial films has been studied at room temperature. There is observed a linear part $J \sim U$ corresponding to Ohm's law and quadratic part $J \sim U^2$ in VAC (fig. 4). $Cd_{1-x}Mn_xTe$ epitaxial films had p-type conductivity with a resistivity of ρ =14.4 · 10⁷ Ohm·cm.

The effect of γ -irradiation on VAC of $Cd_{1-x}Mn_xTe$ epitaxial films is studied at T=300K at doses $D_{\gamma} \le 1.5$ kGy. After irradiation of samples

with γ - quanta at a dose of $D_{\gamma} < 100$ Gy, a parallel shift of the curve occurs towards the current decrease in the entire investigated voltage range (fig. 4). The nature of dependence does not change, Ohmic and quadratic parts are observed, the Ohmic part lengthens. The observed character shows that at irradiation of Cd_{1-x}Mn_xTe thin films with small doses, deep levels are formed in the band gap and are captured by electrons, that leads to a decrease in the conductivity. According to [8], when the sample contains traps with the concentration exceeding the concentration of majority carriers, the carriers injected first are captured by the traps, and the carrier concentration in the conduction band is almost unchanged [9]. At irradiation of samples at $D_{\gamma} = 500$ Gy dose Ohmic part in VAC decreases, quadratic part lengthens, the conductivity increases. Such a behavior of VAC can likely be due to the thermal-field ionization of the traps [10] whose concentration is dependent on the irradiation doze. Further irradiation at doses $D_{\gamma} \ge 1.5$ kGy leads to increasing of Ohmic part and decreasing of quadratic conductivity part, so decreases. Significant decreasing of conductivity is explained by increasing of defects concentration and so destruction of the crystal structure. Obtained results satisfy to the results of our previous works [9].

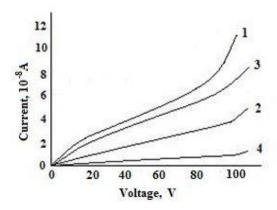


Fig. 4. VAC of irradiated $Cd_{1-x}Mn_xTe$ (x = 0.15) epitaxial films: 1) $D\gamma = 0$, 2) $D\gamma = 100$ Gy, 3) $D\gamma = 500$ Gy, 4) $D\gamma=1.5$ κ Gy

4 Conclusion

Optimal conditions of obtaining perfect $Cd_{1-x}Fe_xSe(x < 0.08)$ and $Cd_{1-x}Mn_xTe(x = 0.15)$ epitaxial films are defined. Crystal structure and surface morphology have been investigated by XRD and SEM methods. XRD studies show that thin films grown on glass substrates at temperature $T_{sub} = 640 \div 670$ K have a monocrystalline and polycrystalline structures. SEM image shows that the obtained Cd_1 -

_xFe_xSe epitaxial films were smooth and glossy, but surface of Cd_{1-x}Mn_xTe epitaxial films were roughness with a grain size of up to 10 microns. The electrical properties of Cd_{1-x}Fe_xSe (x < 0.08) and Cd_{1-x}Mn_xTe (x = 0.15) epitaxial films have been studied at room temperature. It was defined that Cd_{1-x}Fe_xSe SMSC epitaxial films are of n-type and Cd_{1-x}Mn_xTe SMSC epitaxial films are of p-type. It was defined the electrical resistivity 14.4 · 10⁷ Ohm·cm. The effect of γ -irradiation on VAC of Cd_{1-x}Mn_xTe epitaxial films is studied at doses $D_{\gamma} \le 1.5$ κGy.

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Contribution of individual authors to the creation of a scientific article (ghostwriting policy)

Matanat Mehrabova was responsible for the experimental work of electrical properties.

Niyazi Hasanov carried out the XRD and SEM investigations

Vafa Guluzade has executed the experiments of Section 3.

Rafig Sadigov obtained thin films.

Conflicts of Interest

The author(s) declare no potential conflicts of interest concerning the research, authorship, or publication of this article.

Sources of funding for research presented in a scientific article or scientific article itself

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